

Cross-linked Polymers and Rubber Elasticity

Chapter 9 (Sperling)

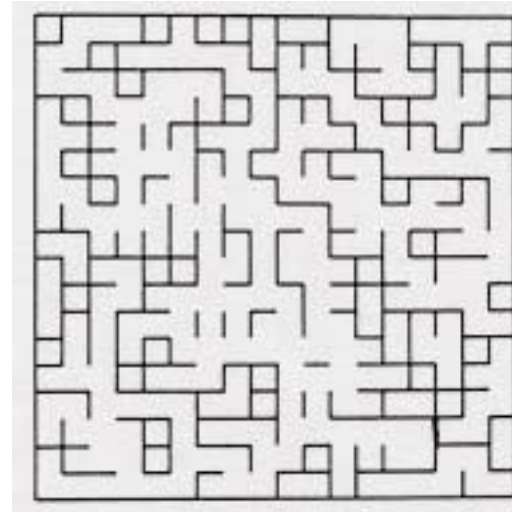
- Definition of Rubber Elasticity and Requirements
- Cross-links, Networks, Classes of Elastomers (sections 1-3, 16)
- Simple Theory of Rubber Elasticity (sections 4-8)
 - Entropic Origin of Elastic Retractive Forces
 - The Ideal Rubber Behavior
- Departures from the Ideal Rubber Behavior (sections 9-11)
 - Non-zero Energy Contribution to the Elastic Retractive Forces
 - Stress-induced Crystallization and Limited Extensibility of Chains (How to make better elastomers: High Strength and High Modulus)
 - Network Defects (dangling chains, loops, trapped entanglements, etc..)
 - Semi-empirical Mooney-Rivlin Treatment (Affine vs Non-Affine Deformation)

Definition of Rubber Elasticity and Requirements

- Definition of Rubber Elasticity:
Very large deformability with complete recoverability.
- Molecular Requirements:
 - **Material must consist of polymer chains.**
Need to change conformation and extension under stress.
 - **Polymer chains must be highly flexible.**
Need to access conformational changes (not w/ glassy, crystalline, stiff mat.)
 - **Polymer chains must be joined in a network structure.**
Need to avoid irreversible chain slippage (permanent strain).
One out of 100 monomers must connect two different chains.
Connections (covalent bond, crystallite, glassy domain in block copolymer)

Cross-links, Networks and Classes of Elastomers

- Chemical Cross-linking Process:
Sol-Gel or Percolation Transition
- Gel Characteristics:
 - Infinite Viscosity
 - Non-zero Modulus
 - One giant Molecule
 - Solid Polymer Network
- Step Polymerization (isocyanates, oxirane)
- Chain Polymerization (styrene with divinylbenzene)
- Post-polymerization Reactions (vulcanization with sulfur)
- Cross-linked and Gels in the Food/Health Care Industry:
Globular protein example: the egg white ! (S_2 bridges)
Jello®, contact lenses, liquid soaps, shampoos (proteins, polysaccharides)



Cross-links, Networks and Classes of Elastomers

- Differentiate Thermosets and Elastomers:
Concerned only with Elastomers (Why ? $T > T_g$)
- Historical Development:
 - Christopher Columbus, 1492
 - Gough, 1805
 - Stretching an elastomer leads to its **warming up**

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Cross-links, Networks and Classes of Elastomers

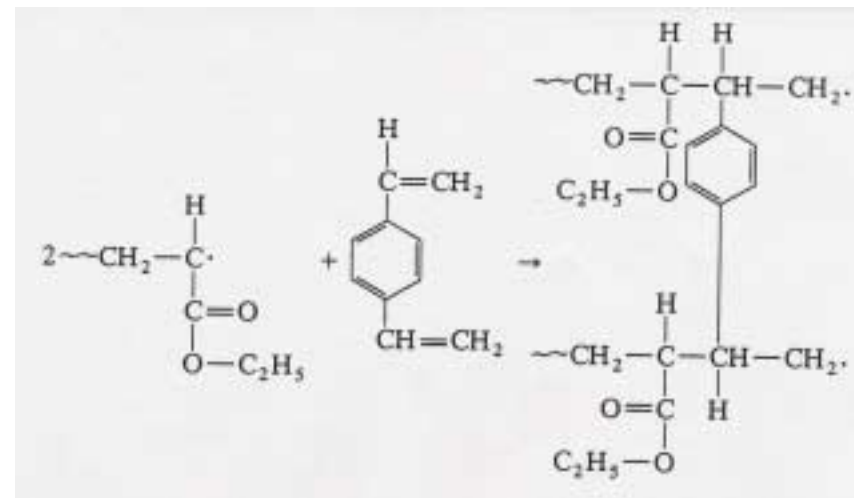
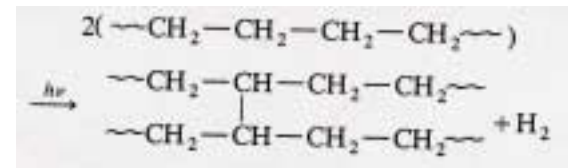
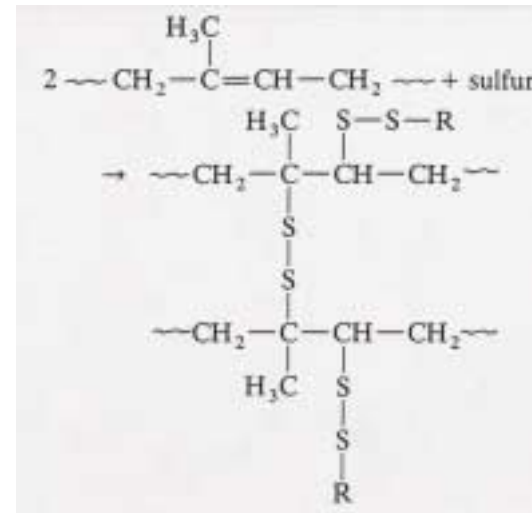
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- Why cross-linking ?
 - Dimensional stability, creep reduction is required in manufactured goods!!!!

Cross-links, Networks and Classes of Elastomers

- Sulfur Vulcanization:
- Electron or γ Irradiation
- Chain Polymerization with a Tetrafunctional Comonomer



Cross-links, Networks and Classes of Elastomers

Name	Structure
A. Diene elastomers	$\left\langle \text{CH}_2 - \overset{\text{X}}{\text{C}} = \text{CH} - \text{CH}_2 \right\rangle_n$
Polybutadiene	X — = H —
Polyisoprene	X — = CH ₃ —
Polychloroprene	X — = Cl —
B. Acrylics	$\left\langle \text{CH}_2 - \overset{\text{O}=\text{C}-\text{O}-\text{X}}{\text{CH}} \right\rangle_n$
Poly(ethyl acrylate)	X — = CH ₃ CH ₂ —
C. EPDM ^a	$\left\langle \text{CH}_2 - \text{CH}_2 \right\rangle_n \left\langle \overset{\text{CH}_3}{\text{CH}} - \text{CH}_2 \right\rangle_m$
D. Thermoplastic elastomers	ABA
Poly(styrene- <i>block</i> -butadiene- <i>block</i> -styrene)	A = polystyrene B = polybutadiene
Segmented polyurethanes	$\left\langle \text{AB} \right\rangle_n$ A = polyether (soft block) B = aromatic urethane (hard block) A = poly(butylene oxide) B = poly(terephthalic acid-ethylene glycol)
E. Inorganic elastomers	
Silicone rubber	$\left\langle \overset{\text{CH}_3}{\text{Si}} - \text{O} \right\rangle_n$
Polyphosphazines	$\left\langle \text{N} = \overset{\text{R}}{\text{P}} \right\rangle_n$

Commercial Products

NBR, SBR

Butyl Rubber & EPDM
ca. add 2% of diene.

Block and Segmented Copolymers

Kraton[®], Lycra[®],
Spandex[®], Hytrel[®]

Caulking, gaskets,
O-rings, high T elastomers

Cross-links, Networks and Classes of Elastomers

Table 9.7 Poly(ether-ester) material characterization (130)

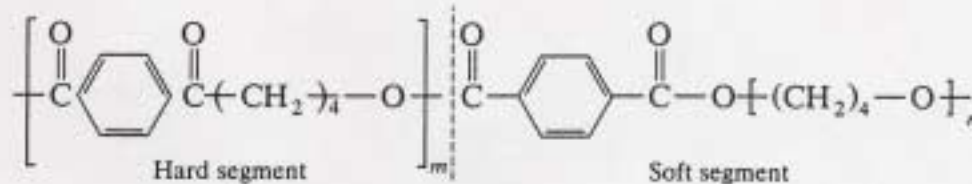
Hard Block (%)	H33	H50	H57	H63	H76	H84
4GT content, wt%	33	50	57	63	76	84
Av 4GT ^a block length, <i>L</i>	2.64	4.95	6.43	8.14	14.8	24.26
DSC						
<i>T_g</i> , °C	-68	-59	-55	-51	-33	-9
<i>T_f</i> , °C	163	189	196	200	209	214
ΔH_f , J/g	16	33	41	48	58	61
Crystallinity %	11.5	22.9	28.6	33.3	40.7	42.8
Young's modulus, <i>E_y</i> , MPa	36.8	102	—	132	328	375
Rheovibron ^b						
<i>E''</i> peak, °C ^c	-63	-58	-53	-48	-30	-4
tan δ peak, °C ^c	-51	-41	-34	-27	10	30

^aTetramethylene terephthalate.

^bCompression-molded samples.

^c β Relaxation.

(129–131):



Segmented Copolymers:

Example: Hytrel[®] (DuPont)

Hard segment: $m = 1, 2$

Soft segment: $n = 40-60$

Cross-links, Networks and Classes of Elastomers

Table 9.8 Typical tire tread recipes (12)

Ingredient	phr ^a		Function
	Natural Rubber	SBR Synthetic	
Smoked sheet	100	—	Elastomer
Styrene-butadiene/oil masterbatch	—	103.1	Elastomer-extender masterbatch
Cis-polybutadiene	—	25	Special purpose elastomer
Oil-soluble sulfonic acid	2.0	5.0	Processing aid
Stearic acid	2.5	2.0	Accelerator-activator
Zinc oxide	3.5	3.0	Accelerator-activator
Phenyl- β -naphthylamine	2.0	2.0	Antioxidant
Substituted <i>N,N'</i> - <i>p</i> -phenylenediamine	4.0	4.0	Antiozonant
Microcrystalline wax	1.0	1.0	Processing aid and finish
Mixed process oil	5.0	7.0	Softener
HAF carbon black	50	—	Reinforcing filler
ISAF carbon black	—	65	Reinforcing filler
Sulfur	2.5	2.8	Vulcanizing agent
Substituted benzothiazole-2-sulfonamide	0.5	1.5	Accelerator
<i>N</i> -nitrosodiphenylamine	0.5	—	Retarder
Total weight	173.5	220.4	
Specific gravity	1.12	1.13	

^aParts per hundred parts of rubber, by weight.

Theory of Rubber Elasticity

- Thermodynamics, Statistical Mechanics and Chain Statistics (Kelvin, Kuhn, Guth, Mark, James, Treloar and Flory)
- Second Law of Thermodynamics:
 - **What happens if you release the stress applied to a cross-linked coil?**
 - A1: Deformation is a Constant Volume Process ($v \approx 1/2$)
 - A2: Helmholtz Free Energy ($F = A = U - TS$) should be minimized
 - A3: Assume the Internal Energy (U) is independent of deformation
 - **Final Answer: the Entropy must be maximized....**
- **IN PLAIN LANGUAGE:**
 - Releasing the stress applied to a cross-linked rubber, leads to the coiling back of extended chains, since a retraction allows the chains to adopt higher entropy conformations.
 - **RETRACTIVE FORCES ARE ENTROPIC IN NATURE IN RUBBERY MATERIALS... What about metals ?**

Theory of Rubber Elasticity

- Approach and Results:
 - Consider the Gaussian statistics for the distribution of end to end distances.
 - Calculate S , the entropy per chain in terms of $\langle r_0^2 \rangle$.
 - Calculate ΔF , the free energy change due to deformation ($-T\Delta S$).
 - Assume local and macroscopic deformations are the same (affine deformation).
 - Assume the volume does not change during deformation.
 - **Rubbery materials are non-Hookean**

n = # active chains per unit volume
 α : elongation ratio

$$F = U - TS = U - kT \ln \Omega(r, T)$$

$$f = \left(\frac{\partial F}{\partial r} \right)_{T, V} = -kT \left(\frac{\partial \ln \Omega(r, T)}{\partial r} \right)_{V, T} = \frac{3kTr}{\bar{r}_0^2}$$

$$\Delta F = -W = \frac{3nRT}{\bar{r}_0^2} \int_{\bar{r}_i}^{\bar{r}^2} r dr = \frac{nRT}{2} \frac{\bar{r}_i^2}{\bar{r}_0^2} \left(\alpha^2 + \frac{2}{\alpha} - 3 \right)$$

$$\sigma = \left(\frac{\partial \Delta F}{\partial \alpha} \right) = nRT \frac{\bar{r}_i^2}{\bar{r}_0^2} \left(\alpha - \frac{1}{\alpha^2} \right)$$

$$E = 3nRT \frac{\bar{r}_i^2}{\bar{r}_0^2}$$

$$G = \frac{E}{2(1+\nu)} = nRT \frac{\bar{r}_i^2}{\bar{r}_0^2}$$

$$\sigma = G \left(\alpha - \frac{1}{\alpha^2} \right)$$

**Ideal Rubber
Behavior**

Theory of Rubber Elasticity

Comparison of Experiment and Theory

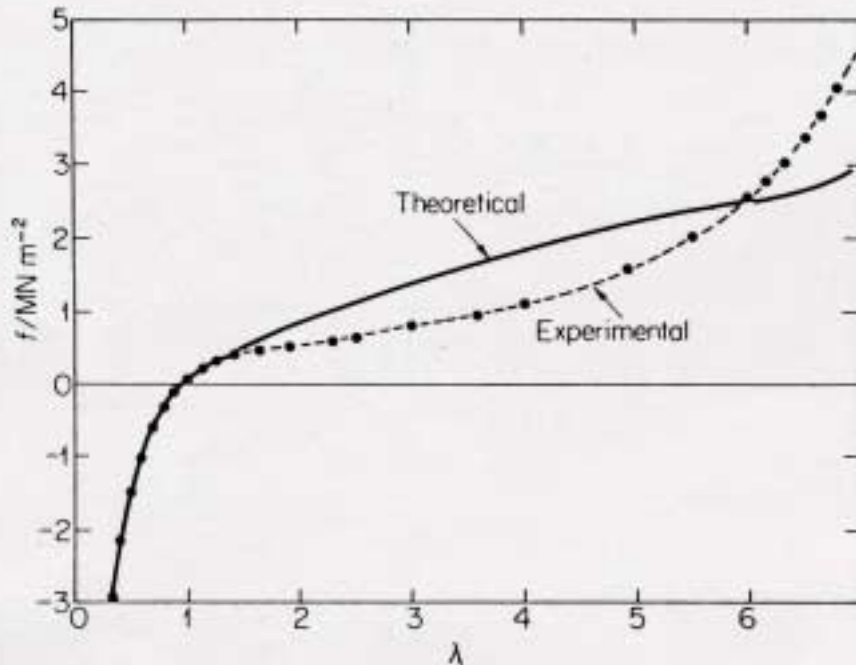
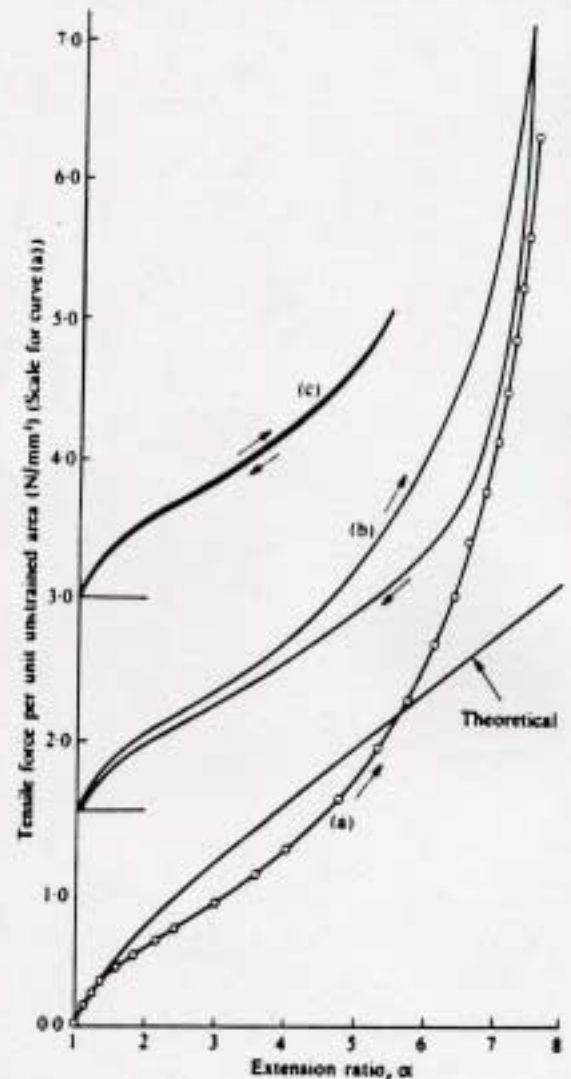


FIGURE 14.6. Extension or compression ratio λ as a function of the tensile or compressive force f for a rubber vulcanizate. Theoretical curve is derived from equation (14.21) using $G = 0.392 \text{ MN m}^{-2}$. (From data by Treloar, 1944.)

Figure 9.5 Stress-strain behavior of lightly cross-linked natural rubber at 50°C . Curve (a), experimental. Theoretical is equation (9.4). Curve (c) illustrates the reversible nature of the extension up to $\alpha = 5.5$. At higher elongations, curve (b), hysteresis effects become important. The theoretical curve has been fitted to the experimental data in the region of small extensions, with $nRT = 0.39 \text{ N / mm}^2$ (37, 38).



Theory of Rubber Elasticity

Typical Calculation

Assume an elastomer of $0.1 \times 0.1 \times 10 \text{ cm}^3$ is stretched to 25 cm length at 35°C , a stress of $2 \times 10^7 \text{ dynes/cm}^2$ being required.

What is the concentration of active network chains and what is the magnitude of the shear modulus?

$$n = \frac{\sigma}{RT \left(\alpha - \frac{1}{\alpha^2} \right)}$$

$$G = nRT$$

$$n = 3.3 \times 10^{-4} \text{ mol/cm}^3 \quad G = 0.86 \text{ J/cm}^3 \text{ or } G = 0.86 \text{ MPa}$$

Theory of Rubber Elasticity

Ideal Rubber Equation of State

$$dU = dw + dq$$

$$dU = -pdV + fdL + TdS = fdL + TdS$$

$$F = U - TS$$

$$dF = dU - TdS - SdT$$

$$dF = fdL - SdT \Rightarrow f = \left(\frac{\partial F}{\partial L} \right)_{V,T}$$

$$f = \left(\frac{\partial F}{\partial L} \right)_{V,T} = \left(\frac{\partial U}{\partial L} \right)_{V,T} - T \left(\frac{\partial S}{\partial L} \right)_{V,T} - \cancel{S \left(\frac{\partial T}{\partial L} \right)_{V,T}}$$

$$dF = fdL - SdT \Rightarrow \left(\frac{\partial S}{\partial L} \right)_{V,T} = - \left(\frac{\partial f}{\partial T} \right)_{V,L}$$

$$f = \left(\frac{\partial U}{\partial L} \right)_{V,T} + T \left(\frac{\partial f}{\partial T} \right)_{V,L}$$

$$f_e = \left(\frac{\partial U}{\partial L} \right)_{V,T}$$

$$f_s = T \left(\frac{\partial f}{\partial T} \right)_{V,L} = -T \left(\frac{\partial S}{\partial L} \right)_{V,T}$$

Ideal Rubber EoS

$$f = T \left(\frac{\partial f}{\partial T} \right)_{V,L}$$

$$f_e = \left(\frac{\partial U}{\partial L} \right)_{V,T} = 0$$

Theory of Rubber Elasticity

Rubber Thermoelastic Behavior

Stress required to achieve a given strain at a fixed temperature

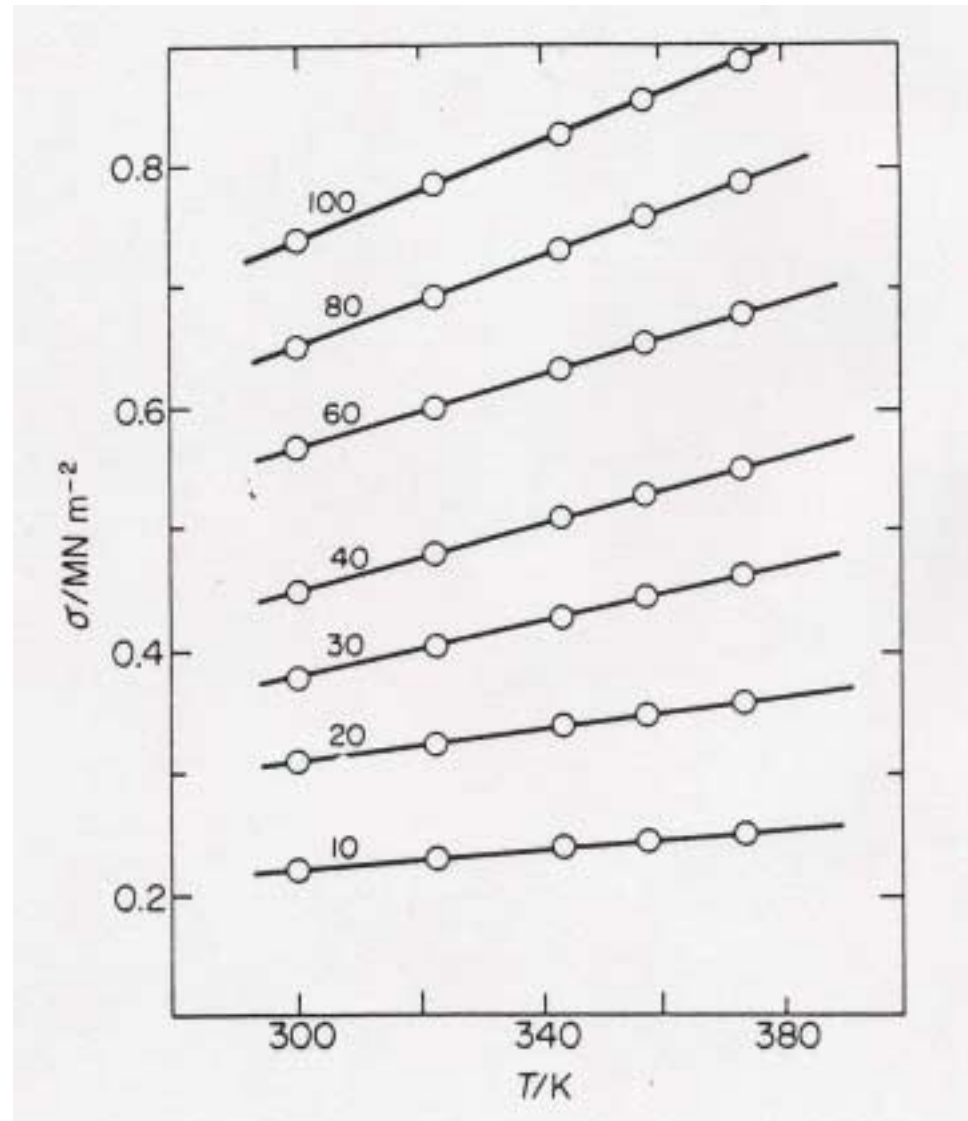
Rubber sample at different percentage strains.

$$\sigma = nRT \frac{\bar{r}_i^2}{\bar{r}_0^2} \left(\alpha - \frac{1}{\alpha^2} \right)$$

$$f = f_e + f_s$$

$$f_s = T \left(\frac{\partial f}{\partial T} \right)_{L,V}$$

$$f_e = \left(\frac{\partial U}{\partial L} \right)_{T,V} = f - T \left(\frac{\partial f}{\partial T} \right)_{L,V}$$



Theory of Rubber Elasticity

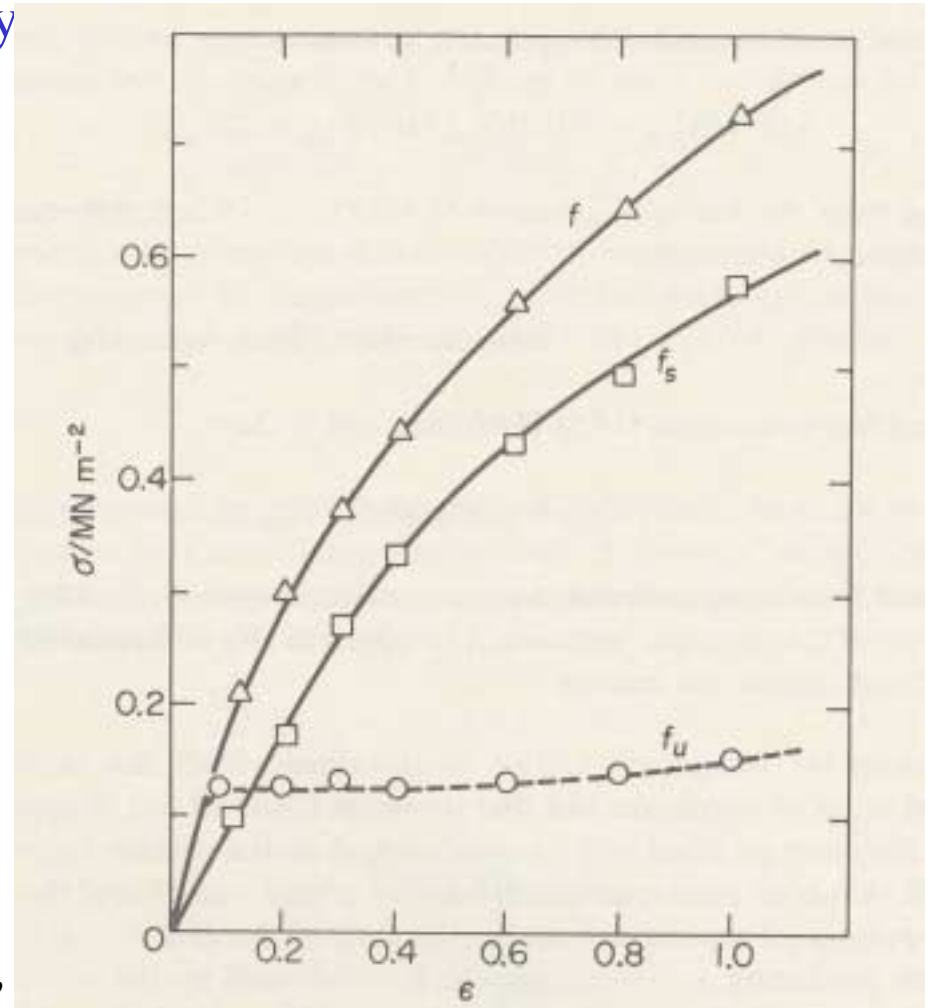
- Is the behavior of a real elastomer truly ideal ?
- How do we explain the observation made by Gough (strain leads to a rise in temperature)?

$$f = f_e + f_s$$

$$f_s = T \left(\frac{\partial f}{\partial T} \right)_{L,V}$$

$$f_e = \left(\frac{\partial U}{\partial L} \right)_{T,V} = f - T \left(\frac{\partial f}{\partial T} \right)_{L,V}$$

$$\left(\frac{\partial T}{\partial L} \right)_{S,p} = - \left(\frac{\partial T}{\partial S} \right)_{L,p} \left(\frac{\partial S}{\partial L} \right)_{T,p} = \frac{T}{C_{p,L}} \left(\frac{\partial f}{\partial T} \right)_{L,p}$$



Departure from Ideal Rubber Behavior

- Departures from the Ideal Rubber Behavior
 - Non-zero Energy Contribution to Elastic Retractive Forces (previous slide)
 - Stress-induced Crystallization
 - Limited Extensibility of Chains
(better elastomers with high strength and high modulus can be obtained when bimodal networks with long and short chain segments are used)
 - Network Defects
They must be accounted for in the calculation of the modulus.
How do we get the true “n” ?

Departure from Ideal Rubber Behavior

- Non-zero Energy Contribution to Elastic Retractive Forces

$$f_e = \left(\frac{\partial U}{\partial L} \right)_{T,V} = f - T \left(\frac{\partial f}{\partial T} \right)_{L,V}$$

$$\frac{f_e}{f} = \frac{1}{T} \frac{d \ln \langle r_o^2 \rangle}{dT}$$

$$[\eta] = \phi \left(\frac{\langle r_o^2 \rangle}{M} \right)^{3/2} M^a$$

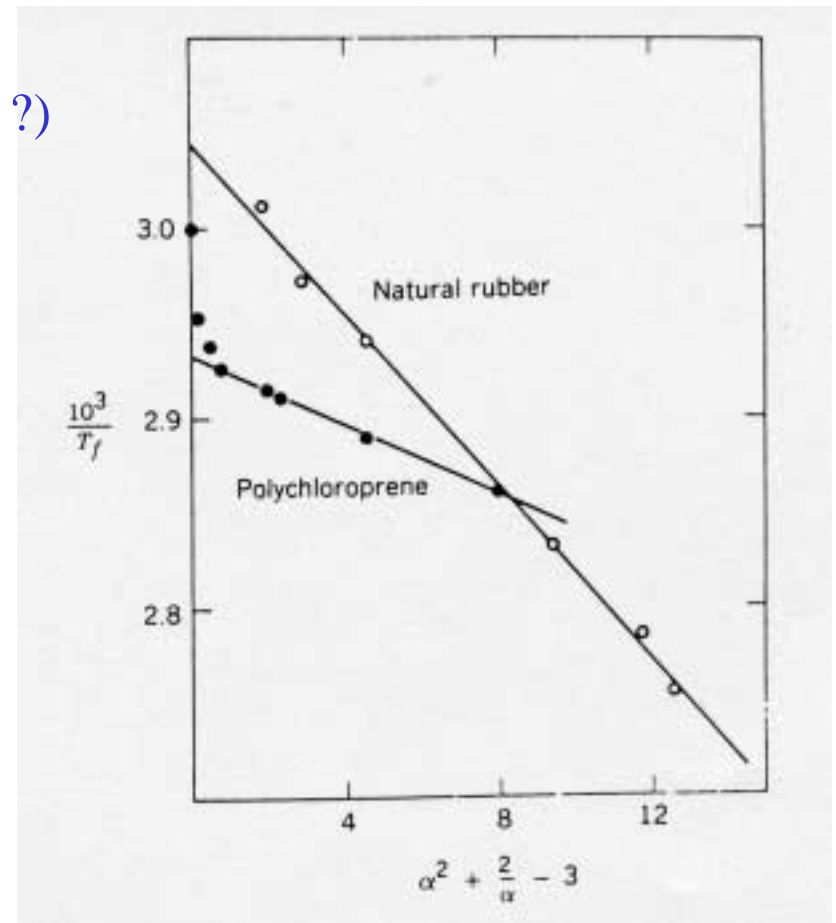
$$\frac{d \ln \langle r_o^2 \rangle}{dT} = \frac{2}{3} \frac{d \ln [\eta]}{dT}$$

Polymer	f_e / f
natural rubber	0.12
t-poly(isoprene)	0.10
cis poly(butadiene)	0.17
poly(ethylene)	-0.42
poly(ethyl acrylate)	-0.16
poly(dimethyl siloxane)	0.15

Departure from Ideal Rubber Behavior

- Stress-Induced Crystallization
 - One of the two possible reasons for up-swing of stress at high strains
 - Modulus increases rapidly as crystals act as new cross-links !
 - Only for crystallizable chains !
 - Melting point increases with stress (why ?)

$$\frac{1}{T_m} = \frac{1}{T_m^0} - \frac{R}{2N_u \Delta H_f^0} \left(\alpha^2 + \frac{2}{\alpha} - 3 \right)$$



Departure from Ideal Rubber Behavior

- Limited Chain Extensibility

- As chains extend significantly, the Gaussian Statistics approximation no longer holds.
- Mooney-Rivlin-Saunders semi-empirical theory allows to extend the predicted stress-strain curves from low to moderate extensions (not high extensions, $\alpha < 2.5$ see fig. 9-18).
- How can we make better elastomers (high strengths, high modulus) ?
- $C_1 = 2 - 6 \text{ kg/cm}^2$, $C_2 = 2 \text{ kg/cm}^2$
- $2C_2/C_1$ measures looseness of network (Flory)

$$\sigma = 2 \left(C_1 + \frac{C_2}{\alpha} \right) \left(\alpha - \frac{1}{\alpha^2} \right)$$

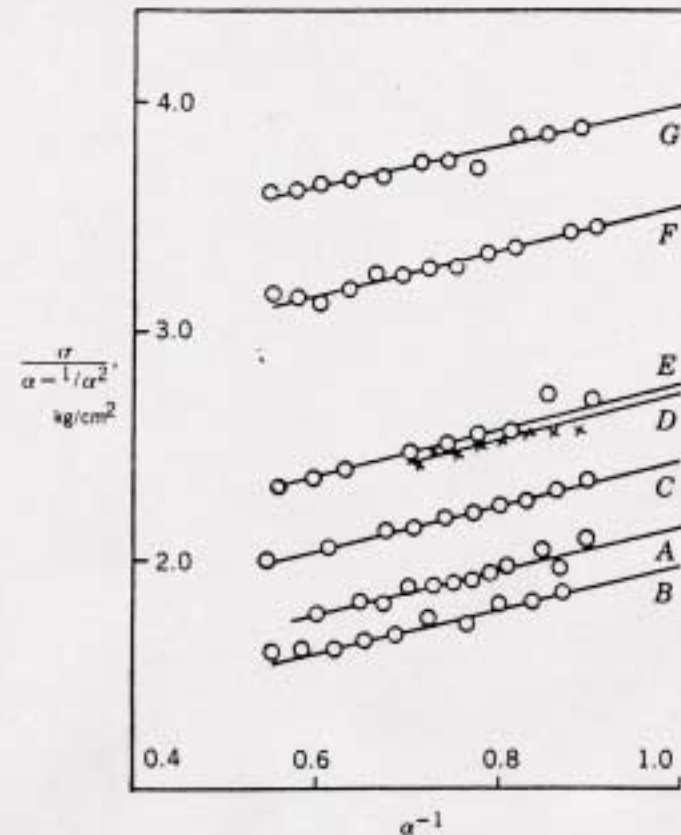
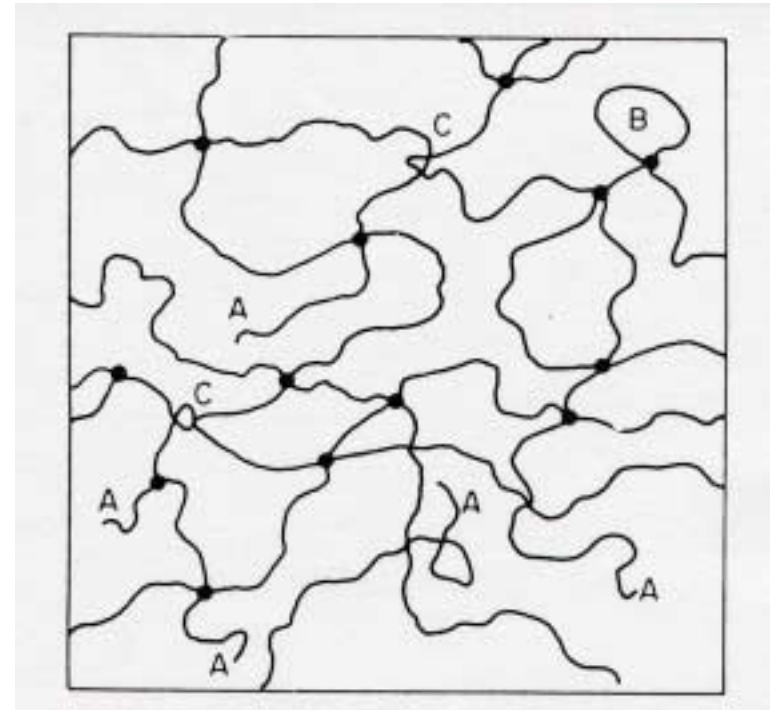


Figure 9.15 Plot of $\sigma / (\alpha - 1/\alpha^2)$ versus α^{-1} for a range of natural rubber vulcanizates. Sulfur content increases from 3 to 4%, with time of vulcanization and other quantities as variables (48).

Departure from Ideal Rubber Behavior

- Network Defects
 - Loose Chain Ends (A.), Intramolecular Loops (B.), Entangled Chain Loops (C.)
 - The quantity n in the Rubber Elasticity Theory is the true density of active chain sections between cross-links. It could be calculated from stress-strain curves.
 - This number can also be estimated from the extent of swelling. The larger n , the lower the extent of swelling. Swelling is the isotropic expansion of the network as solvent molecules diffuse in the network.

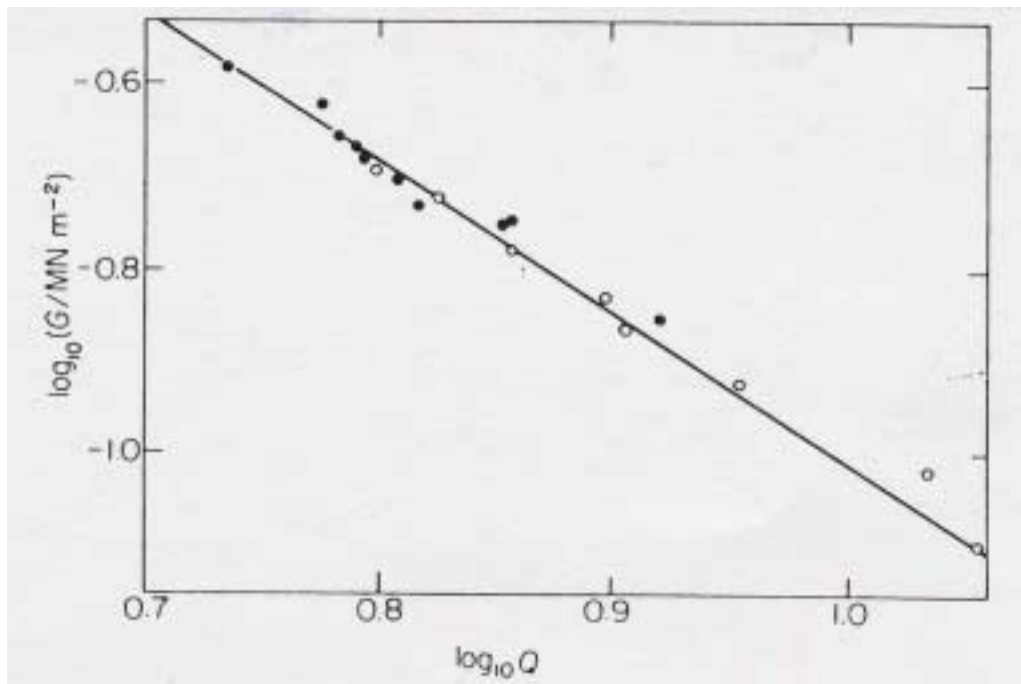


$$\sigma = \left[n_c \left(1 - \frac{2M}{M_c} \right) + n_p \right] RT \frac{\bar{r}_i^2}{\bar{r}_0^2} \left(\alpha - \frac{1}{\alpha^2} \right)$$

Accounts qualitatively for the effect of chain ends and entanglements

Swelling of Elastomers

Density of Active Network Chains
Swelling vs. Shear Modulus



Flory-Rehner Equation

$$\Delta G_{\text{SWEL}} = \Delta G_{\text{MIX}} + \Delta G_{\text{ELAST}}$$

$$\Delta G_{\text{elas}} = \frac{3}{2} nRT \left[\frac{1}{v_2^{2/3}} - 1 \right]$$
$$-\left(\ln[1 - v_2] + \chi_1 v_2^2 + v_2 \right) = nV_1 \left(v_2^{1/3} - \frac{v_2}{2} \right)$$

Flory-Rehner and
Rubber Elasticity
Theories

A: Constant
Q: Swelling Ratio

$$G = \frac{ART}{V_1 Q^{5/3}}$$
$$Q = \frac{V}{V_0} = \frac{1}{v_2}$$